## X-RAY DIFFRACTION (XRD)

and
XRAY POWDER DIFFRACTION (XRPD)
Structure Determination by use of X-Rays
Determination of sites/positions of atoms
Crystal or X-Ray Structure Determination
Finger print, Identification of Substances
Phase Analysis, Phase Transition Investigation

## Production of X-rays by X-ray tubes



## Production of X-rays by X-ray tubes



X-ray tube
Anode of e.g. $\mathrm{Cr}, \mathrm{Cu}, \mathrm{Mo}$ )

: Schaltschema eines Röntgengenerators (nach Jost, 1975).


## Pount focus

X-ray tube
Scheme, Anode, Focus

## White/Slow and Emission Spectra of X-rays


$\lambda_{\text {Min }}[\AA]=12.4 / \mathrm{V}[\mathrm{kV}], \lambda_{\text {Max }} \approx 1.5 \cdot \lambda_{\text {Min }}$


White and emission spectra (Mo)

$$
\nu_{\mathrm{K} \alpha} \sim \mathrm{Z}^{2}(\text { Henry Moseley, } 1913)
$$

## Emission Spectrum of a X-ray Tube



Slow and emission spectra
$v_{\mathrm{Ka}} \sim \mathrm{Z}^{2}$ (Henry Moseley, 1913)
$\lambda_{\text {Min }}[\AA]=12.4 / \mathrm{V}[\mathrm{kV}], \lambda_{\text {Max }} \approx 1.5 \cdot \lambda_{\text {Min }}$
$\log E$

## K Seria




Term scheme (Auswahlregeln: $2 \mathrm{n}-1$ states, $1 \leq \mathrm{n} \leq 7,0 \leq 1 \leq \mathrm{n}-1, \Delta \mathrm{l}= \pm 1,-1 \leq \mathrm{m}_{1} \leq+1, \Delta \mathrm{~m}_{1}=0, \pm 1$

## Wavelengths of different X-ray tubes

Wavelengths of the most important K series in $\AA$ *

| Atom Nr. | Element | $\mathrm{K} \alpha$ | $\mathrm{K} \alpha_{1}$ | $\mathrm{~K} \alpha_{2}$ | $\mathrm{~K} \beta$ |
| :---: | :--- | :---: | :---: | :---: | :---: |
| 24 | Chrom | 2,29092 | 2,28962 | 2,29351 | 2,08480 |
| 26 | Eisen | 1,93728 | 1,93597 | 1,93991 | 1,75653 |
| 27 | Kobalt | 1,79021 | 1,78892 | 1,79278 | 1,62075 |
| 28 | Nickel | 1,65912 | 1,65784 | 1,66169 | 1,50010 |
| 29 | Kupfer | 1,54178 | 1,54051 | 1,54433 | 1,39217 |
| 42 | Molybdän | 0,71069 | 0,70926 | 0.71354 | 0.63225 |
| 47 | Silber | 0,56083 | 0,55936 | 0,56378 | 0,49701 |
| 74 | Wolfram | 0,21060 | 0,20899 | 0,21381 | 0,18436 |

* $1 \AA=10^{-10} \mathrm{~m}$. In former times wavelengths were given in X units:
$1000 \mathrm{X}=1 \mathrm{KX}=1,00202 \AA=100,202 \mathrm{pm}$.


## Radiation Protection and Units for Use of X-Ray's

| Activity: Ion dosis | Becquerel (Bq) (C/kg): | $1 \mathrm{~Bq}=1 / \mathrm{s}$ | before Curie ( Ci ): $1 \mathrm{Ci}=3,7 \times 10^{10} \mathrm{~Bq}$ <br> before Röntgen ( R ): $1 \mathrm{R}=2,6 \times 10^{-4} \mathrm{C} / \mathrm{kg}$ ) |
| :---: | :---: | :---: | :---: |
| Energy dosis: | Gray (Gy) | $1 \mathrm{~Gy}=1 \mathrm{~J} / \mathrm{kg}$ | before $\operatorname{Rad}(\mathrm{rd}): 1 \mathrm{rd}=0,01 \mathrm{~Gy}$ |
| Eqivalent dosis: | Sievert (Sv) | $1 \mathrm{~Sv}=1 \mathrm{~J} / \mathrm{kg}$ | before Rem (rem): $1 \mathrm{rem}=0,01 \mathrm{~Sv}$ |
| - Activity: | 1 Ci is the decay rate of $1 \mathrm{~g}{ }^{226} \mathrm{Ra}$ (historically to the honor of Marie Curie). |  |  |
| - Iondosis: | physically, referred to the building of ions in air. |  |  |
| - Energy dosis: | absorbed radiation energy per mass unit. |  |  |
| - Eqivalent dosis | : measure for destruction ability of a radiaton (impact factor • energy dosis). |  |  |
| Impact factor: | 1 for X-rays, | to 20 for $\alpha$ |  |

- Natural radiation exposure: $\sim 2,5 \mathrm{mSv} / \mathrm{a}$ (cosmisc $\sim 1 \mathrm{mSv}$, terrestric $\sim 1 \mathrm{mSv}$, other $\sim 0,5 \mathrm{mSv}$ ), ${ }^{40} \mathrm{~K}$ corporated in a human body gives e.g. $\sim 4500 \mathrm{~Bq}(\sim 0,18 \mathrm{mSv} / \mathrm{a})$.
- Medical radiation exposure: $\sim 1,5 \mathrm{mSv} / \mathrm{a}$ (e.g. stomach-bowel investigations $\sim 160 \mathrm{mSv}$ ).
- Other radiation exposures (technique, atomic bombes, nuclear reactors etc.): $\sim 0.01-0.03 \mathrm{mSv} / \mathrm{a}$.
- 20 mSv have been fixed to be the maximum body dosis per year for exposed persons.
- 400 mSv per year are considered to be just not harmful. From 2-10 Sv serious health damages appeare. Doses of $10-15 \mathrm{~Sv}$ are lethal by $90-100 \%$, doses $>50 \mathrm{~Sv}$ are lethal by $100 \%$ within 1 h to 2 days.


## $\longrightarrow$ Use of ionizing radiation needs attention and shielding

Literature: Hans Kiefer, Winfried Koelzer: Strahlen und Strahlenschutz, Springer-Verlag
Internet: www.bfs.de (Bundesamt für Strahlenschutz)

## Radiation Protection Needs Shielding



X-ray tube



X-ray tube with shielding

## X-Ray Detectors



Classical detector: Geiger-Müller Counter

## X-Ray Detectors

Scintillation counter


Modern detector: Szintillation counter

## X-Ray Detectors



Modern site-sensitive Detector
One or two dimensional detection of angle or direction of X-rays are diffracted to

- Spectrum: Absorption edges


## Absorption of X-Rays

- Reduction by absorber


$$
I=I_{0} \cdot e^{-\alpha d}
$$

- Absorptionskoeffizient $\alpha$
- Halbwertsdicke

$$
d_{1}=\alpha^{-1} \cdot \ln (2)
$$

Absorption of X-rays


Filtering effects of a Ni-Foil for $\mathrm{Cu}-\mathrm{K}_{\alpha}$ radiation (monochromating) ${ }^{12}$

## Scattering/Diffraction of X-rays

If a substance is irradiated by electromagnetic Radiation or neutrons of suitable wavelengths, a small part of the primary radiation $\left(\sim 10^{-6}\right)$ is scattered by the electrons or nuclei of the atoms or ions or molecules of the sample elastically $(\Delta \mathrm{E}=0)$ and coherently ( $\Delta \varphi=$ konstant $)$ in all directions. The resulting scattering/diffraction pattern $\mathbf{R}$ is the Fourier transform of the electron/scattering distribution function $\boldsymbol{\rho}$ of the sample and vice versa.


The shape of the resulting scattering/diffraction pattern depends on the degree of order of the sample.

## A. X-ray Scattering Diagram of an Amorphous Sample

no long-range order, no short range order
$\mathrm{I}(\theta)$
(monoatomic gas e.g. He) $\Rightarrow$ monotoneous decrease


$$
\mathrm{I}(\theta)=\mathrm{N} \cdot \mathrm{f}^{2}
$$

$$
\mathrm{f}=\text { scattering length of atoms } \mathrm{N}
$$

$\Rightarrow$ no information

## $\mathrm{I}(\theta)$

no long-range, but short range order


## B. X-ray Scattering Diagram of a Crystalline Sample

crystals and crystal powders have long-range and short-range order $\Rightarrow$ discontinious scattering diagrams with discrete reflections

crystal powder
orientation statistical, $\lambda$ fixed
$\Rightarrow$ cones of interference

Debye-Scherrer diagram

single crystal
orientation or $\lambda$ variable
$\Rightarrow$ dots of interference (reflections)

## Atoms in crystals are three-dimensionally ordered


forming lattice plane families (Miller indices hkl, spacings $\mathrm{d}_{\mathrm{hkl}}$ )

## Diffraction of X-rays (neutrons, electrons!) by a Crystalline Sample

 (Single Crystal or Crystal Powder)X-rays scattered by a crystalline sample are not totally extinct only for those directions, where the scattered rays are ,in phase". $\mathbf{R}(\mathbf{S})$ und $\mathrm{I}(\theta)$ therefore are periodic functions of „Bragg reflections".


Bragg equation: $\mathrm{n} \cdot \lambda=2 \mathrm{~d} \cdot \sin \theta$ or $\lambda=2 \mathrm{~d}_{(\mathrm{hkl})} \cdot \sin \theta_{(\mathrm{hkl})}$

## Diffraction of X-rays by crystalline samples



Directions and planes of a regular lattice with Miller indices hkl and spacings $\mathrm{d}_{\mathrm{hkl}}$


Bragg equation:
$\mathrm{n} \lambda=2 \mathrm{~d} \sin \theta$
$\lambda=2 \mathrm{~d}_{(\mathrm{hkl})} \sin \theta_{(\mathrm{hkl})}$

## X-ray Diffraction (XRD)



The crystal or crystallite is positioned at the center of a (virtual) sphere of radius $1 / \lambda$ and is hit by a X-ray beam with wave length $\lambda$ running along a center line of that sphere.

## X-ray Diffraction (XRD)

## Reciprocal lattice

## X-ray Powder Diffraction (XRPD)

## (Ewald sphere)

$$
\lambda=2 \mathrm{~d}_{\mathrm{hk} 1} \cdot \sin \theta_{\mathrm{hkl}}
$$

primary beam $\mathbf{s}_{0}$ with $\left|\mathbf{s}_{0}\right|=1 / \lambda$

Theorem of Thales Theorem of Pythagoras


The origin of the reciprocal lattice, combined with the crystal(ite), is shifted along the X-ray beam (primary beam $\mathbf{s}_{0}$ ) to the circumference of the sphere (by $\mathbf{s}_{0}$ ). Then the Bragg equation is fulfilled, always if the scattering vector $\mathbf{S}=\mathbf{s}-\mathbf{s}_{0}$ is equal to a reciprocal lattice vector $\mathbf{H}_{\mathrm{hkl}}=$ ha*+kb*+lc* , i.e. if $\mathbf{S}=\mathbf{H}$, means, if a reciprocal lattice point matches the Ewald sphere.

## (X-ray) Diffraction of a Crystalline Sample

## (Single Crystal or Crystal Powder) detector



Fourier transform of the electron density distribution


V : volume of sample $\overrightarrow{\mathrm{r}}$ : vector in space R : scattering amplitude

## X-ray Powder Diffraction (XRPD)

Superposition (interference) of the Xrays scattered by the electrons of the atoms results in enhancement (a) or extinction (b) of the X-rays.

- X-rays scattered by an atom are described by the atomic scattering or form factor $\mathrm{f}_{\mathrm{j}}$.
- X-rays scattered by all atoms of a unit cell of a crystal are described by the structure factor $\mathbf{F}_{\text {hkl }}$.

b) Extinction (out of phase)

Superposition (interference) of X-rays

## X-ray Powder Diffraction (XRPD)



Scattering strength $f_{j 0}$ of a non-vibrating single atom (atomic form factor, atomic scattering factor) as a function of $\sin \theta / \lambda$

## X-ray Powder Diffraction (XRPD)

## Structure factor $\mathbf{F}_{\mathrm{hkl}}$

The scattering power of all atoms of an unit cell of a crystal is characterized by the so called structure factor $\mathbf{F}_{\mathrm{hkl}}$. It is (for $\theta=0$ ) proportional to the sum of the scattering contributions of all the atoms of the unit cell. $\mathbf{F}_{\text {hkl }}$ is characteristic for every family of lattice planes (hkl) and in general a complex number. In a unit cell with n atoms, the structure factor is:

$$
F_{h k 1}=\sum_{j=1}^{n} f_{j} \exp \left[2 \pi i\left(h x_{j}+k y_{j}+1 z_{j}\right)\right]
$$

$\mathrm{h}, \mathrm{k}, \mathrm{l}$ : Miller indices, $\mathrm{x}_{\mathrm{j}}, \mathrm{y}_{\mathrm{j}}, \mathrm{z}_{\mathrm{j}}$ : atomic positional coordinates.

Using the Euler equation $\exp (\mathrm{i} \varphi)=\cos \varphi+\mathrm{i} \sin \varphi$, the structure factor becomes:
$\mathbf{F}_{\mathrm{hkl}}=\sum \mathrm{f}_{\mathrm{j}} \cos 2 \pi\left(\mathrm{hx}_{\mathrm{j}}+\mathrm{ky}_{\mathrm{j}}+\mathrm{lz}_{\mathrm{j}}\right)+\mathrm{i} \sum \mathrm{f}_{\mathrm{j}} \sin 2 \pi\left(\mathrm{hx}_{\mathrm{j}}+\mathrm{ky}_{\mathrm{j}}+\mathrm{lz}_{\mathrm{j}}\right)$


Measurable is only the intensity, i.e. the square of the structure amplitude: $\mathbf{I}_{\mathbf{h k l}} \sim \mathbf{F}_{\mathbf{h k l}}{ }^{\mathbf{r}}$ This means that all the phases of the complex numbers $\mathbf{F}_{\mathrm{hkl} 1}$ (or the signs in case of centrosymmetric crystal structures) are lost.
三 „Phase problem of crystal structure analysis/determination"

## X-ray Powder Diffraction (XRPD)

## Structure factor $\mathbf{F}_{\mathrm{hk} 1}$

If the structure has a center of symmetry (centrosymmetric structure), the structure factor

$$
\mathbf{F}_{\mathrm{hkl}}=\sum \mathrm{f}_{\mathrm{j}} \cos 2 \pi\left(\mathrm{hx}_{\mathrm{j}}+\mathrm{ky}_{\mathrm{j}}+\mathrm{lz}_{\mathrm{j}}\right)+\mathrm{i} \sum \mathrm{f}_{\mathrm{j}} \sin 2 \pi\left(\mathrm{hx}_{\mathrm{j}}+\mathrm{ky}_{\mathrm{j}}+\mathrm{lz}_{\mathrm{j}}\right)
$$

reduces/simplifies by compensation/elimination of the imaginary parts to

$$
\mathbf{F}_{\mathrm{hkl}}=\sum \mathrm{f}_{\mathrm{j}} \cos 2 \pi\left(\mathrm{hx}_{\mathrm{j}}+\mathrm{ky}_{\mathrm{j}}+\mathrm{lz}_{\mathrm{j}}\right)
$$

thus the „phase problem" reduces to a „sign problem".
Structure amplitude $\mathrm{F}_{\mathrm{hkl}}=\left|\mathrm{F}_{\mathrm{hkl}}\right|$ and Scattering intensity $\mathrm{I}_{\mathrm{hkl}}$
The modulus of the structure factor is named scattering or structure amplitude.
The scattering intensity ist proportional to the square of the structure amplitude: $\mathbf{I}_{\mathbf{h k l}} \sim\left|\mathbf{F}_{\mathbf{h k l}}\right|^{2}$.
The structure amplitudes can be calculated (after correction for absorption, extinction, and Lorentz-polarisation effects) from the intensities $\mathrm{I}_{\mathrm{hkl}}(\rightarrow$ data reduction):

$$
\mathbf{I}_{\mathrm{hkl}}=\mathbf{K} \cdot \mathbf{F} \cdot \mathbf{A} \cdot \mathbf{E} \cdot \mathbf{L} \mathbf{p} \cdot\left|\mathbf{F}_{\mathrm{hk}}\right| 2
$$

( $\mathrm{K}=$ scale factor, $\mathrm{F}=$ coincidence factor, $\mathrm{A}=$ absorption factor, $\mathrm{E}=$ extinction factor,
$\mathrm{Lp}=$ Lorentz-polarisation factor)

## X-ray Powder Diffraction (XRPD)

In a powder sample all crystallites are statistically (randomly) oriented. Thus a powder sample produces for each family of lattice planes hkl a distinct scattering cone of high intensity


The cone angle is $\mathbf{4 \theta _ { \mathrm { hkl } }}$ ( 4 x the scattering angle $\theta_{\mathrm{hkl}}$ )
With the scattering angle $\boldsymbol{\theta}_{\mathrm{hk}}$, the lattice plane distance $\mathbf{d}_{\mathrm{hk} l}$ of the respective family of lattice planes can be calculated by use of the Bragg equation ( $\lambda=$ wave length):

$$
\mathrm{d}_{\mathrm{hkl}}=\lambda /\left(2 \sin \theta_{\mathrm{hkl}}\right) .
$$

## X-ray Powder Diffraction (XRPD)

Diffraction cones (reflections) with randomly or symmetry-caused identical d values fall together leading to symmetry-caused coincidences $\rightarrow$ Net plane occurence factor

Flächenhäufigkeitsfaktoren für Pulververfahren

| $h k l$ | kubisch | tetrag. | hexagonal | rhombisch | monoklin | triklin |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: |
| $h k l$ | 48 | 16 | 24 | 8 | 4 | 2 |
| $h h l$ | 24 | 8 | 12 | 8 | 4 | 2 |
| $h l h$ | 24 | 16 | 24 | 8 | 4 | 2 |
| $l h h$ | 24 | 16 | 24 | 8 | 4 | 2 |
| $h k 0$ | 24 | 8 | 12 | 4 | 2 | 2 |
| $h 0 l$ | 24 | 16 | 12 | 4 | 4 | 2 |
| $0 h l$ | 24 | 16 | 12 | 4 | 4 | 2 |
| $h h h$ | 8 | 8 | 12 | 8 | 4 | 2 |
| $h h 0$ | 12 | 4 | 6 | 4 | 2 | 2 |
| $h 0 h$ | 12 | 8 | 12 | 4 | 4 | 2 |
| $0 h h$ | 12 | 8 | 12 | 4 | 4 | 2 |
| $h 00$ | 6 | 4 | 6 | 2 | 2 | 2 |
| $0 k 0$ | 6 | 4 | 6 | 2 | 2 | 2 |
| $00 l$ | 6 | 2 | 2 | 2 | 2 | 2 |

## X-ray Powder Diffraction (XRPD)

## Debye-Scherrer geometry



Debye-Scherrer pattern using a flat film


Debye-Scherrer pattern using a cylindric film

## X-ray Powder Diffraction (XRPD)

Debye-Scherrer geometry


$$
\leftarrow---------180^{\circ} \geq 2 \theta_{\text {hkl }} \geq 0^{\circ}-------\rightarrow
$$



$$
\leftarrow 360-4 \theta_{\mathrm{hkl}} \rightarrow
$$

$$
\leftarrow 4 \theta_{h k l} \rightarrow
$$

## X-ray Powder Diffraction (XRPD)

## X-ray powder diffractometer



X-ray powder diffractometer and scattering geometry of/in a sample

## X-ray Powder Diffraction (XRPD)



X-ray diffraction pattern of a powder sample

## X-ray Powder Diffraction (XRPD)



## Powder diffractometer with Bragg-Brentano geometry

Normal of the sample bisects the angle between primary and diffracted beam directions.

- Sample is fixed, tube and detector turn to each other by an angle $\theta$.
- Tube is fixed, sample and detector turn by an angle $\theta$, and $2 \theta$, respectively in the same direction.


## X-ray Powder Diffraction (XRPD)



Beam course for the Bragg-Brentano geometry

## X-ray Powder Diffraction (XRPD)



Powder Diffractometer Bruker AXS D 5000

## X-ray Powder Diffraction (XRPD)



Schematic diagram of the beam course in a powder diffractometer with a focusing monochromator and PS detektor

## X-ray Powder Diffraction (XRPD)



Standard measurement in Bragg-Brentano geometry (corundum plate )

## X-ray Powder Diffraction (XRPD)



Comparison with own data file

| d | 3.99 | 2.91 | 4.73 | 9.47 | $\mathrm{Ba}\left(\mathrm{N}_{3}\right)_{2}$ |  |  |  |  |  | \% |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $1 / 1_{1}$ | 100 | 80 | 70 | 20 | Baritm Azide |  |  |  |  |  |  |
| Rad. <br> Filter <br> Dia. <br> Cut off $\square$ <br> I/I $I_{1}$ Visual <br> Ref. Tokar, Krischner and Rad1, Montash. Chem, 96, 3, 932-40 (1965) |  |  |  |  |  | d A | 1/11 | hk1 | d A | I/ $\mathrm{I}_{1}$ | hkl |
|  |  |  |  |  |  | 9.47 | 20 | 001 | 2.295 | 40 | 211 |
|  |  |  |  |  |  | 5.33 | 20 | 100 | 2.282 | 40 | 210 |
|  |  |  |  |  |  | 5.03 | 60 | 101 | 2.196 | 70 | 020,113 |
| $\begin{aligned} & \text { Sys. Monoclinic } \\ & \mathbf{a}_{0} 5.42 \quad b_{0}{ }^{4.39} \\ & a \\ & \text { Ref. Ibid. } \end{aligned}$ |  |  |  |  | $\stackrel{C}{C} \underset{D \times 3.25}{ }$ | 4.73 4.35 | 70 30 | 002 101 | 2.182 2.152 | 20 10 | ${ }_{211}^{202}$ |
|  |  |  | 3.99 |  |  | 100 | 011 | 2.083 | 30 | 014 |
|  |  |  | 3.39 |  |  | 60 | 110 | 2.046 | 10 | 118 |
|  |  |  | 3.27 |  |  | 20 | 102 | 2.030 | 10 | 120 |
| $\begin{aligned} & \epsilon a \\ & 2 \mathrm{~V} \\ & \text { Ref. Ibid. } \end{aligned}$ |  | $\begin{gathered} \mathrm{n} \omega \beta \\ 3.22 \end{gathered}$ |  | mp | Color ${ }^{\text {Sign }}$ |  | 3.15 | 30 | 003 | 2.013 | 40 | 121 |
|  |  | 3.08 |  |  |  |  | 70 | 111 | 1.992 | 60 | 213,022 |
|  |  | 2.943 |  |  |  |  | 20 | 103 | 1.961 | 30 | 121 |
|  |  | 2.909 | 80 |  |  |  | 112 | 1.952 | 30 | 212 |
|  |  | $\begin{aligned} & 2.693 \\ & 2.673 \end{aligned}$ | 40 |  |  |  | 201 | 1.942 | 10 | 204 |
|  |  |  |  |  |  | 30 | 200 | 1.893 | 20 | 005,203 |
|  |  |  |  |  |  | 2.627 | 10 | 112 | 1.823 | 20 | 122 |
|  |  |  |  |  |  | 2.562 2.537 | 70 10 | 013 | 1,803 | 30 | 023 |
|  |  |  |  |  |  | 2.514 | 10 | 202 |  |  |  |
|  |  |  |  |  |  | 2.468 | 20 | 201 |  |  |  |
|  |  |  |  |  |  | 2.366 | 10 | 004 |  |  |  |

Comparison with JCPDS

Phase analysis or identification of a sample using XRPD (JCPDF = Joint Commitee of Powder Diffraction File)

## X-ray Powder Diffraction (XRPD)



Illustration of a phase change by use of XRPD patterns

## X-ray Powder Diffraction (XRPD)



- D8 ADVANCE,
- Cu radiation, 40kVI40 mA
- Divergence aperture: 0,1
- Increment: 0.007

Counting time/step: 0.1 sec
Speed: $4.2^{\circ} / \mathrm{min}$.
Total time: 3:35 min.

Small angle scattering of silver behenate $\left(\mathrm{CH}_{3}\left(\mathrm{CH}_{2}\right)_{20}\right.$ - COOAg$)$
(Bragg-Brentano geometry)

## X-ray Powder Diffraction (XRPD)



Quantitative phase analysis of cement

## X-ray Powder Diffraction (XRPD)



Quantitative phase analysis of cement

## Literature

-Röntgenfeinstrukturanalyse von H. Krischner, Vieweg (Allgemeine Einführung, Schwerpunkt Pulvermethoden) oder alternativ
-Röntgen-Pulverdiffraktometrie von Rudolf Allmann, Clausthaler
Tektonische Hefte 29, Sven von Loga, 1994
-Kristallstrukturbestimmung von W. Massa, Teubner, Stuttgart, 1984
-Untersuchungsmethoden in der Chemie von H. Naumer und W. Heller, Wiley-VCH
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-X-Ray Structure Determination von G. H. Stout, L.H. Jensen, MacMillan, London
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