X-RAY DIFFRACTION (XRD) and XRAY POWDER DIFFRACTION (XRPD)

Structure Determination by use of X-Rays Determination of sites/positions of atoms Crystal or X-Ray Structure Determination Finger print, Identification of Substances Phase Analysis, Phase Transition Investigation

Production of X-rays by X-ray tubes





Ionisation by fast electrons → Emission of X-rays

X-ray tube Anode (e.g. Cr, Cu, Mo)

Production of X-rays by X-ray tubes





: Schaltschema eines Röntgengenerators (nach Jost, 1975).



Pount focus

X-ray tube Anode of e.g. Cr, Cu, Mo) X-ray tube Scheme, Anode, Focus

White/Slow and Emission Spectra of X-rays



Emission Spectrum of a X-ray Tube



Slow and emission spectra $v_{K\alpha} \sim Z^2$ (Henry Moseley, 1913) λ_{Min} [Å] = 12.4/V [kV], $\lambda_{Max} \approx 1.5 \cdot \lambda_{Min}$



Term scheme (Auswahlregeln: 2n-1 states, $1 \le n \le 7, 0 \le l \le n-1, \Delta l = \pm l, -l \le m_l \le +l, \Delta m_l = 0, \pm 1$

Wavelengths of different X-ray tubes

Wavelengths of the most important K series in Å *

Atom Nr.	Element	Κα	Και	Kα ₂	κβ
24	Chrom	2,29092	2,28962	2,29351	2,08480
26	Eisen	1,93728	1,93597	1,93991	1.75653
27	Kobalt	1,79021	1,78892	1,79278	1,62075
28	Nickel	1,65912	1,65784	1,66169	1.50010
29	Kupfer	1,54178	1,54051	1,54433	1,39217
42	Molybdän	0,71069	0,70926	0.71354	0.63225
47	Silber	0,56083	0,55936	0,56378	0.49701
74	Wolfram	0,21060	0,20899	0,21381	0,18436

* 1 Å = 10^{-10} m. In former times wavelengths were given in X units: 1000 X = 1KX = 1,00202 Å = 100,202 pm.

Radiation Protection and Units for Use of X-Ray's

Activity:	Becquerel (Bq)	1 Bq = 1/s	before Curie (Ci): $1Ci = 3,7x10^{10}$ Bq
Ion dosis	(C/kg):		before Röntgen (R): $1 R = 2.6 \times 10^{-4} C/kg$)
Energy dosis:	Gray (Gy)	1 Gy = 1 J/kg	before Rad (rd): $1 \text{ rd} = 0,01 \text{ Gy}$
Eqivalent dosis	: Sievert (Sv)	1 Sv = 1 J/kg	before Rem (rem): $1 \text{ rem} = 0,01 \text{ Sv}$

- Activity: 1 Ci is the decay rate of 1 g 226 Ra (historically to the honor of Marie Curie).
- Iondosis: physically, referred to the building of ions in air.
- Energy dosis: absorbed radiation energy per mass unit.
- Eqivalent dosis: measure for destruction ability of a radiaton (impact factor \cdot energy dosis). Impact factor: 1 for X-rays, up to 20 for α -rays.
- Natural radiation exposure: ~2,5 mSv/a (cosmisc ~1 mSv, terrestric ~1 mSv, other ~0,5 mSv), 40 K corporated in a human body gives e.g. ~4500 Bq (~0,18 mSv/a).
- Medical radiation exposure: \sim 1,5 mSv/a (e.g. stomach-bowel investigations \sim 160 mSv).
- Other radiation exposures (technique, atomic bombes, nuclear reactors etc.): ~0.01-0.03 mSv/a.
- 20 mSv have been fixed to be the maximum body dosis per year for exposed persons.
- 400 mSv per year are considered to be just not harmful. From 2-10 Sv serious health damages appeare. Doses of 10-15 Sv are lethal by 90-100 %, doses >50 Sv are lethal by 100 % within 1h to 2 days.

\longrightarrow Use of ionizing radiation needs attention and shielding

Literature: Hans Kiefer, Winfried Koelzer: Strahlen und Strahlenschutz, Springer-Verlag Internet: <u>www.bfs.de</u> (Bundesamt für Strahlenschutz)

Radiation Protection Needs Shielding



X-ray tube



X-ray tube with shielding

X-Ray Detectors



Classical detector: Geiger-Müller Counter

X-Ray Detectors

Scintillation counter



Modern detector: Szintillation counter

X-Ray Detectors



Modern site-sensitive Detector

One or two dimensional detection of angle or direction of X-rays are diffracted to

• Spectrum: Absorption edges

Absorption of X-Rays

• Reduction by absorber



- Absorptionskoeffizient α
- Halbwertsdicke

$$d_{\frac{1}{2}} = \alpha^{-1} \cdot \ln(2)$$

Absorption of X-rays



Filtering effects of a Ni-Foil for Cu- K_{α} radiation (monochromating) ¹²

Scattering/Diffraction of X-rays

If a substance is irradiated by electromagnetic Radiation or neutrons of suitable wavelengths, a small part of the primary radiation (~ 10⁻⁶) is scattered by the electrons or nuclei of the atoms or ions or molecules of the sample elastically ($\Delta E = 0$) and coherently ($\Delta \phi = \text{konstant}$) in all directions. The resulting scattering/diffraction pattern **R** is the **Fourier transform** of the electron/scattering distribution function ρ of the sample and vice versa.

sample

$$\rho(\vec{r}) = \frac{\int_{V} \rho(\vec{r}) \exp(2\pi i \vec{r} \cdot \vec{S}) dV}{\rho(\vec{r}) = \frac{1}{V} \int_{V^*} R(\vec{S}) \exp(-2\pi i \vec{r} \cdot \vec{S}) dV^*} \qquad \text{diffr. pattern} R(\vec{S})$$

The shape of the resulting scattering/diffraction pattern depends on the degree of order of the sample.

A. X-ray Scattering Diagram of an Amorphous Sample



B. X-ray Scattering Diagram of a Crystalline Sample



Atoms in crystals are three-dimensionally ordered



forming lattice plane families (Miller indices hkl, spacings d_{hkl})

Diffraction of X-rays (neutrons, electrons!) by a Crystalline Sample (Single Crystal or Crystal Powder)

X-rays scattered by a crystalline sample are <u>not totally extinct</u> only for those directions, where the <u>scattered rays are , in phase</u>. **R**(**S**) und I(θ) therefore are periodic functions of ,,Bragg reflections".



Bragg equation: $n \cdot \lambda = 2d \cdot \sin\theta$ or $\lambda = 2d_{(hkl)} \cdot \sin\theta_{(hkl)}$

Diffraction of X-rays by crystalline samples



Directions and planes of a regular lattice with Miller indices hkl and spacings d_{hkl}



X-ray Diffraction (XRD)



The crystal or crystallite is positioned at the center of a (virtual) sphere of radius $1/\lambda$ and is hit by a X-ray beam with wave length λ running along a center line of that sphere.

X-ray Diffraction (XRD)

Reciprocal lattice



The origin of the reciprocal lattice, combined with the crystal(ite), is shifted along the X-ray beam (primary beam s_0) to the circumference of the sphere (by s_0). Then the Bragg equation is fulfilled, always if the scattering vector $S = s - s_0$ is equal to a reciprocal lattice vector $\mathbf{H}_{hkl} = h\mathbf{a}^{*}+k\mathbf{b}^{*}+l\mathbf{c}^{*}$, i.e. if $\mathbf{S} = \mathbf{H}$, means, if a reciprocal lattice point matches the Ewald sphere.

(X-ray) Diffraction of a Crystalline Sample

(Single Crystal or Crystal Powder) detector



Fourier transform of the electron density distribution





: scattering vector \equiv vector in Fourier (momentum) space

S

Superposition (interference) of the Xrays scattered by the **electrons of the atoms** results in enhancement (a) or extinction (b) of the X-rays.

- X-rays scattered by an atom are described by the **atomic scattering** or form factor f_i .

- X-rays scattered by all atoms of a unit cell of a crystal are described by the **structure factor** \mathbf{F}_{hkl} .



Superposition (interference) of X-rays



Scattering strength f_{j0} of a non-vibrating single atom (atomic form factor, atomic scattering factor) as a function of $\sin\theta/\lambda$

Structure factor F_{hkl}

The scattering power of **all** atoms of an unit cell of a crystal is characterized by the so called **structure factor** \mathbf{F}_{hkl} . It is (for $\theta = 0$) proportional to the sum of the scattering contributions of all the atoms of the unit cell. \mathbf{F}_{hkl} is characteristic for every family of lattice planes (hkl) and in general a complex number. In a unit cell with n atoms, the structure factor is:

 $F_{hkl} = \sum_{j=1}^{n} f_j \exp [2\pi i (hx_j + ky_j + lz_j)]$

h,k,l: Miller indices, x_i , y_j , z_j : atomic positional coordinates.

Using the **Euler equation** $exp(i\phi) = cos\phi + i sin\phi$, the structure factor becomes:

$$\mathbf{F}_{hkl} = \sum f_j \cos 2\pi (hx_j + ky_j + lz_j) + i \sum f_j \sin 2\pi (hx_j + ky_j + lz_j)$$



Measurable is only the intensity, i.e. the square of the structure amplitude: $I_{hkl} \sim F_{hkl}^2$. This means that all the phases of the complex numbers F_{hkl} (or the signs in case of centrosymmetric crystal structures) are lost.

≡ "Phase problem of crystal structure analysis/determination"

Structure factor F_{hkl}

If the structure has a center of symmetry (centrosymmetric structure), the structure factor

 $\mathbf{F}_{hkl} = \sum f_j \cos 2\pi (hx_j + ky_j + lz_j) + i \sum f_j \sin 2\pi (hx_j + ky_j + lz_j)$ reduces/simplifies by compensation/elimination of the imaginary parts to

 $\mathbf{F}_{hkl} = \sum f_j \cos 2\pi (hx_j + ky_j + lz_j),$

thus the "**phase problem**" reduces to a "**sign problem**".

Structure amplitude $F_{hkl} = |F_{hkl}|$ and Scattering intensity I_{hkl} The modulus of the structure factor is named scattering or structure amplitude. The scattering intensity ist proportional to the square of the structure amplitude: $I_{hkl} \sim |F_{hkl}|^2$. The structure amplitudes can be calculated (after correction for absorption, extinction, and Lorentz-polarisation effects) from the intensities I_{hkl} (\rightarrow data reduction):

$$\mathbf{I}_{hkl} = \mathbf{K} \cdot \mathbf{F} \cdot \mathbf{A} \cdot \mathbf{E} \cdot \mathbf{L} \mathbf{p} \cdot |\mathbf{F}_{hkl}| \mathbf{2}$$

(K = scale factor, F = coincidence factor, A = absorption factor, E = extinction factor, Lp = Lorentz-polarisation factor)

In a powder sample all crystallites are statistically (randomly) oriented. Thus a powder sample produces for each family of lattice planes **hkl** a distinct scattering cone of high intensity



The cone angle is $4\theta_{hkl}$ (4 x the scattering angle θ_{hkl})

With the scattering angle θ_{hkl} , the lattice plane distance d_{hkl} of the respective family of lattice planes can be calculated by use of the Bragg equation (λ = wave length):

$$d_{hkl} = \lambda / (2 \sin \theta_{hkl}).$$

Diffraction cones (reflections) with randomly or symmetry-caused identical d values fall together leading to symmetry-caused coincidences \rightarrow Net plane occurence factor

hkl	kubisch	tetrag.	hexagonal	rhombisch	monoklin	triklin
hkl	48	16	24	8	4	2
hhl	24	8	12	8	4	2
hlh	24	16	24	8	4	2
lhh	24	16	24	8	4	2
hk 0	24	8	12	4	2	2
h0l	24	16	12	4	4	2
0 <i>kl</i>	24	16	12	4	4	2
hhh	8	8	12	8	4	2
hh0	12	4	6	4	2	2
h0h	12	8	12	4	4	2
0hh	12	8	12	4	4	2
h00	6	4	6	2	2	2
0k0	6	4	6	2	2	2
001	6	2	2	2	2	2

Flächenhäufigkeitsfaktoren für Pulververfahren

Debye-Scherrer geometry



Debye-Scherrer pattern using a flat film Debye-Scherrer pattern using a cylindric film

Debye-Scherrer geometry



X-ray powder diffractometer



X-ray powder diffractometer and scattering geometry of/in a sample

X-ray powder diffractometer



X-ray diffraction pattern of a powder sample



Powder diffractometer with Bragg-Brentano geometry

Normal of the sample bisects the angle between primary and diffracted beam directions.

- Sample is fixed, tube and detector turn to each other by an angle θ .
- Tube is fixed, sample and detector turn by an angle θ , and 2θ , respectively in the same direction.



Beam course for the Bragg-Brentano geometry



Powder Diffractometer Bruker AXS D 5000



Schematic diagram of the beam course in a powder diffractometer with a focusing monochromator and PS detektor 36



Standard measurement in Bragg-Brentano geometry (corundum plate)



Comparison with own data file

4.73 9.47 Ba(N3)2 70 20 Barium Azide d A 1/11 hk1 d A 1/1 hkl Filter Dia. 9.47 20 001 2,295 40 211 I/I1 Visual Ref. Tokar, Krischner and Radl, Montash. Chem. 96, 3, 5.33 20 2.282 100 40 210 60 70 30 5.03 101 2.196 70 020,113 4.73 002 2.182 20 202 SG 4.35 101 2.152 10 211 co 9.59 C 100 3.99 011 2,083 30 014 Z 2 Dx 3.25 3.39 60 110 2.046 10 114 3.27 20 2.030 10 120 121 102 30 40 3.15 003 2,013 3.08 70 60 213,022 111 1.992 Sign 20 Color 2.943 103 1,961 30 121 mp 80 30 10 2.909 112 1,952 212 204 40 2,693 20I 1,942 2.673 30 200 1.893 20 005,203 2.627 10 1.823 112 20 122 2.562 70 013 1,803 30 023 2.537 10 103 10 202 2.514 2.468 20 10 201 2.366 004

Comparison with JCPDS

Phase analysis or identification of a sample using XRPD (JCPDF = Joint Commitee of Powder Diffraction File)



Illustration of a phase change by use of XRPD patterns

Small angle scattering of silver behenate $(CH_3(CH_2)_{20}$ -COOAg) (Bragg-Brentano geometry)

Quantitative phase analysis of cement

Quantitative phase analysis of cement

Literature

•*Röntgenfeinstrukturanalyse* von **H. Krischner**, Vieweg (Allgemeine Einführung, Schwerpunkt Pulvermethoden) oder alternativ

•*Röntgen-Pulverdiffraktometrie* von **Rudolf Allmann**, Clausthaler Tektonische Hefte 29, Sven von Loga, 1994

•Kristallstrukturbestimmung von W. Massa, Teubner, Stuttgart, 1984

•Untersuchungsmethoden in der Chemie von H. Naumer und W. Heller, Wiley-VCH

(Einführung in die moderne Analytik und Strukturbestimmungsmethoden)

•*X-Ray Structure Determination* von **G. H. Stout, L.H. Jensen,** MacMillan, London

(Einführung in die Kristallstrukturanalyse für Fortgeschrittene)